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Redetermination of AgNb₂PS₁₀ revealing a silver deficiency

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Key indicators: single-crystal X-ray study; T = 290 K; mean $\sigma(S-P) = 0.004 \text{ Å}$; disorder in main residue; R factor = 0.045; wR factor = 0.109; data-to-parameter ratio = 16.5.

In comparison with a previous crystallographic study [Goh et al. (2002). J. Solid State Chem. 168, 119-125] of the title compound, silver diniobium tris(disulfide) tetrathiophosphate(V), that reports a full occupation of the silver position and isotropic displacement parameters for the atoms, the current redetermination reveals a silver deficiency with a site-occupation factor of 0.88 (1) and reports all atoms with anisotropic displacement parameters. The structure of $Ag_{0.88}Nb_2PS_{10}$ is composed of ∞ [Nb₂PS₁₀] chains, which are built up from pairs of distorted bicapped trigonal-prismatic [NbS₈] polyhedra forming [Nb₂S₁₂] dimers and of tetrahedral [PS₄] groups. These chains are connected via the statistically disordered Ag⁺ ions, forming double layers. Adjacent layers are stacked solely through van der Waals forces into a threedimensional structure. Short and long Nb-Nb distances [2.880 (1) and 3.770 (2) Å, respectively] alternate along the chain and S_2^{2-} and S_2^{2-} anionic species are observed.

Related literature

The synthesis and structural characterization of stoichiometric AgNb₂PS₁₀ and NaNb₂PS₁₀ have been published (Goh *et al.*, 2002). For Nb₂PS₁₀-related quaternary thiophosphates with general formula *M*Nb₂PS₁₀, see: Do & Yun (1996) for KNb₂PS₁₀, Kim & Yun (2002) for RbNb₂PS₁₀, Kwak *et al.* (2007) for CsNb₂PS₁₀, and Bang *et al.* (2008) for TlNb₂PS₁₀; for related pentanary thiophosphates *M,M*′Nb₂PS₁₀, see: Kwak & Yun (2008) for K_{0.34}Cu_{0.5}Nb₂PS₁₀, Dong *et al.* (2005*a*) for K_{0.5}Ag_{0.5}Nb₂PS₁₀, and Dong *et al.* (2005*b*) for Rb_{0.38}Ag_{0.5}Nb₂PS₁₀. For data standardization, see: Gelato & Parthé (1987). For ionic radii, see: Shannon (1976). For structure validation, see: Spek (2009). For typical P—S bond distances, see: Brec *et al.* (1983). For typical Nb⁴⁺—Nb⁴⁺ bond distances, see: Angenault *et al.* (2000).

Experimental

Crystal data

$Ag_{0.88}Nb_2PS_{10}$	$V = 2408.6 (9) \text{ Å}^3$
$M_r = 631.78$	Z = 8
Monoclinic, C2/c	Mo $K\alpha$ radiation
a = 24.001 (5) Å	$\mu = 5.1 \text{ mm}^{-1}$
b = 7.7711 (17) Å	T = 290 K
c = 12.960 (3) Å	$0.60 \times 0.06 \times 0.04 \text{ mm}$
$\beta = 94.833 \ (19)^{\circ}$	

Data collection

MAC Science MXC3 diffractometer Absorption correction: analytical (de Meulenaer & Tompa, 1965) $T_{\rm min} = 0.727, \, T_{\rm max} = 0.821$ every 100 reflections $T_{\rm min} = 0.727, \, T_{\rm max} = 0.821$ intensity decay: none 2114 independent reflections

Refinement

$R[F^2 > 2\sigma(F^2)] = 0.045$	128 parameters
$wR(F^2) = 0.109$	$\Delta \rho_{\text{max}} = 1.82 \text{ e Å}^{-3}$
S = 1.16	$\Delta \rho_{\min} = -1.20 \text{ e Å}^{-3}$
2114 reflections	

Table 1 Selected geometric parameters (\mathring{A} , $^{\circ}$).

2.536 (3)	$Nb1-S10^{iv}$	2.659 (2)
2.620 (3)	Nb2-S3ii	2.476 (3)
2.875 (3)	Nb2-S7	2.479 (3)
2.916 (3)	$Nb2-S5^{ii}$	2.508 (2)
2.965 (4)	$Nb2-S2^{vii}$	2.551 (3)
3.091(3)	$Nb2-S4^{ii}$	2.558 (2)
2.462(2)	Nb2-S6	2.569 (3)
2.466 (2)	Nb2-S9	2.630 (3)
2.518 (3)	Nb2-S10	2.656 (2)
2.551 (2)	$P-S1^{vi}$	2.009 (4)
2.554 (3)	P-S8	2.048 (4)
2.562 (2)	P-S9	2.059 (4)
2.573 (3)	P—S10	2.065 (3)
108.46 (17)	$S1^{vi}-P-S10$	117.65 (16)
112.81 (17)	S8-P-S10	104.24 (14)
111.87 (16)	S9-P-S10	101.46 (14)
	2.620 (3) 2.875 (3) 2.916 (3) 2.916 (3) 2.965 (4) 3.091 (3) 2.462 (2) 2.466 (2) 2.518 (3) 2.551 (2) 2.552 (2) 2.573 (3) 108.46 (17) 112.81 (17)	2.620 (3) Nb2-S3 ⁱⁱ 2.875 (3) Nb2-S7 2.916 (3) Nb2-S5 ⁱⁱ 2.965 (4) Nb2-S2 ^{vii} 3.091 (3) Nb2-S4 ⁱⁱ 2.462 (2) Nb2-S6 2.466 (2) Nb2-S9 2.518 (3) Nb2-S10 2.551 (2) P-S1 ^{vii} 2.552 (2) P-S9 2.573 (3) P-S9 2.573 (3) S1 ^{vi} -P-S10 108.46 (17) S1 ^{vi} -P-S10 112.81 (17) S8-P-S10

Data collection: *MAC Science MXC3* (MAC Science, 1994); cell refinement: *MAC Science MXC3*; data reduction: *MAC Science MXC3*; program(s) used to solve structure: *SHELXS97* (Sheldrick, 2008); program(s) used to refine structure: *SHELXL97* (Sheldrick, 2008); molecular graphics: locally modified version of *ORTEP* (Johnson, 1965); software used to prepare material for publication: *WinGX* (Farrugia, 1999).

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Supplementary data and figures for this paper are available from the IUCr electronic archives (Reference: WM2240).

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Acta Cryst. (2009). E65, i56-i57 [doi:10.1107/S1600536809025100]

Redetermination of AgNb₂PS₁₀ revealing a silver deficiency

J. Do and H. Yun

Comment

In comparison with the previous study of AgNb₂PS₁₀ (Goh *et al.*, 2002), both lattice parameters and atomic coordinates of the current redetermination are the same within their standard deviations. However, our investigation indicated that there is a deficiency of Ag atoms in the title compound with a site occupation factor (s.o.f.) of 0.88 (1). This observation is consistent with crystal structure refinements from crystals obtained from other reaction batches. Therefore we assume that the crystal originally investigated by Goh *et al.* (2002) shows the same behaviour. In general, non-stoichiometry in multinary niobium thiophosphates is not uncommon and has been observed in one of our previous studies (Kwak & Yun, 2008).

The structure of $Ag_{0.88}Nb_2PS_{10}$ consists of one-dimensional $_{\infty}^{1}[Nb_2PS_{10}]$ chains along the [001] direction that are connected *via* the statistically disordered Ag^+ ions to form a double layer parallel to the *bc* plane. These layers then stack on top of each other to form the three-dimensional structure with a van der Waals gap as shown in Fig. 1. There is no bonding interaction, only van der Waals forces, between the double layers.

As shown in other phases in the M,MNb₂PS₁₀ family, viz KNb₂PS₁₀ (Do & Yun, 1996), RbNb₂PS₁₀ (Kim & Yun, 2002), CsNb₂PS₁₀ (Kwak et al., 2007), TlNb₂PS₁₀ (Bang et al., 2008), K_{0.34}Cu_{0.5}Nb₂PS₁₀ (Kwak & Yun, 2008), K_{0.5}Ag_{0.5}Nb₂PS₁₀ (Dong et al., 2005a), and Rb_{0.38}Ag_{0.5}Nb₂PS₁₀ (Dong et al., 2005b), each of the chains is made up of pairs of [NbS₈] polyhedra forming characteristic [Nb₂S₁₂] units and tetrahedral [PS₄] groups. In the title compound, the Nb1 and Nb2 atoms are surrounded by 8 S atoms in a bicapped trigonal-prismatic fashion. Two prisms are sharing a rectangular face to form the [Nb₂S₁₂] unit. This unit shows an approximate 2-fold rotational symmetry and the rotation axis bisects the short Nb1—Nb2 distance and the (S—S)²⁻ sides of the rectangular face shared by each trigonal prism. The [Nb₂S₁₂] unit is bound to each other to form the infinite [Nb₂S₉] chains by sharing the S—S prism edge. One of the S atoms at the prism edge and two other capping S atoms are bound to the P atom and an additional S atom (S1) is attached to the P atom to complete the [PS4] tetrahedral coordination. The P—S distances (2.048 (4)–2.065 (3) Å) are in good agreement with P—S distances found in related phases (Brec et al., 1983). The S1 atom is the only sulfur atom that is not coordinated to any of the Nb atoms causing the short P—S1 distance (2.009 (4) Å) as well as the large ADP of the S1 atom (Do & Yun, 1996). Along the chains, the Nb atoms associate in pairs with Nb—Nb interactions alternating in the sequence of one short and one long distances. Although the short distance (2.880 (1) Å) is typical of Nb⁴⁺—Nb⁴⁺ bonding interactions (Angenault *et al.*. 2000), the long distance (3.770 (2) Å) implies that there is no significant Nb—Nb interaction and such an arrangement is consistent with the highly resistive and diamagnetic nature of the compound.

The silver atom is surrounded by six S atoms. The coordination around the Ag atom can be described as [2 + 4] (Fig. 2). Two S atoms are coordinated to the Ag atom (Ag—S1, 2.536 (3) Å; Ag—S9, 2.620 (3) Å), whereas four S atoms are weakly bound to the Ag atoms (Ag—S, 2.875 (3)–3.091 (3) Å). These distances are comparable to the sum of the ionic radii of each element, 2.51 Å for CN=2 and 2.99 Å for CN=6 (Shannon, 1976).

Experimental

Ag_{0.88}Nb₂PS₁₀ was prepared by the reaction of the elements Nb, P, and S with an elemental ratio of 2:1:10 in the eutectic mixture of AgCl/LiCl (Kojima, 99.5%). The starting materials, Nb powder (CERAC 99.8%), P powder (CERAC 99.5%), and S powder (Aldrich 99.999%) were placed in a silica glass tube. The mass ratio of reactants and halide flux was 1:2. The tube was evacuated to 0.133 Pa, sealed, and heated to 973 K where it was kept for 7d. Afterwards, the tube was cooled at a rate of 4 K/h to room temperature. Black needle-shaped crystals were isolated from the flux by leaching out with water. The crystals are stable in water and air. Electron microprobe analysis of the crystals established their homogeneity and the presence of Ag, Nb, P and S. No other element was detected.

Refinement

Large anisotropic displacement parameters (ADPs) of the silver atom were found when the structure was refined with the stoichiometric model $AgNb_2PS_{10}$, (Goh *et al.*, 2002). The deficient nature of the Ag site was checked by refining the occupancy of Ag while that of the other atoms were fixed. With the non-stoichiometric model ($Ag_xNb_2PS_{10}$), the occupation factor of the Ag site was reduced significantly from 1 to 0.88 (1) and the reliability factor (wR2 = 0.1089) was improved in comparison with full occupation of the silver position (wR2 = 0.1341). In addition, the anisotropic displacement parameters in the disordered model became plausible. As no evidence was found for ordering of the Ag site, a statistically disordered structure was assumed. With the composition established, the data for the compound were corrected for absorption with the use of the analytical method (de Meulenaer & Tompa, 1965). The highest residual electron density is 0.96 Å from the S1 site and the deepest hole is 0.73 Å from the Ag site. No additional symmetry, as tested by *PLATON* (Spek, 2009), was detected in this structure. Structure data were finally standardized by means of the program *STRUCTURE TIDY* (Gelato & Parthé, 1987).

Figures



Fig. 1. A view of the structure of $Ag_{0.88}Nb_2PS_{10}$ down the *b* axis showing the double layers and the two-dimensional nature of the compound. Large and small filled circles are Nb and P atoms respectively; large open circles are S atoms; grey circles represent Ag atoms.

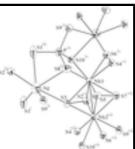


Fig. 2. A view of the structure of $Ag_{0.88}Nb_2PS_{10}$ showing the coordination around Nb, Ag, and P atoms. Anisotropic displacement ellipsoids are drawn at the 70% probability level. Symmetry codes are as given in Table 1.

silver diniobium tris(disulfide) tetrathiophosphate(V)

Crystal data

 $Ag_{0.88}Nb_2PS_{10}$ $F_{000} = 2385$

 $M_r = 631.78$ $D_x = 3.485 \text{ Mg m}^{-3}$

Monoclinic, C2/c Mo $K\alpha$ radiation, $\lambda = 0.71073$ Å Hall symbol: -C 2yc Cell parameters from 24 reflections

a = 24.001 (5) Å $\theta = 10.0-15.0^{\circ}$ b = 7.7711 (17) Å $\mu = 5.1 \text{ mm}^{-1}$ c = 12.960 (3) Å T = 290 K $\beta = 94.833 (19)^{\circ}$ Needle, black

 $V = 2408.6 (9) \text{ Å}^3$ $0.60 \times 0.06 \times 0.04 \text{ mm}$

Z = 8

Data collection

MAC Science MXC3 diffractometer $R_{\text{int}} = 0.017$

Radiation source: normal-focus sealed tube $\theta_{max} = 25.0^{\circ}$ Monochromator: graphite $\theta_{min} = 1.7^{\circ}$

T = 290 K $h = -28 \rightarrow 28$ $\omega - 2\theta \text{ scans}$ $k = 0 \rightarrow 9$

Absorption correction: analytical $l = 0 \rightarrow 15$

(de Meulenaer & Tompa, 1965) $l = 0 \rightarrow 15$

 $T_{\min} = 0.727$, $T_{\max} = 0.821$ 2 standard reflections 2221 measured reflections every 100 reflections 2114 independent reflections intensity decay: none

1835 reflections with $I > 2\sigma(I)$

Refinement

Refinement on F^2 Primary atom site location: structure-invariant direct

methods

Least-squares matrix: full Secondary atom site location: difference Fourier map

 $R[F^2 > 2\sigma(F^2)] = 0.045$ $w = 1/[\sigma^2(F_0^2) + (0.029P)^2 + 98.0899P]$

where $P = (F_0^2 + 2F_c^2)/3$

 $WR(F^2) = 0.109$ $(\Delta/\sigma)_{\text{max}} < 0.001$

S = 1.16 $\Delta \rho_{\text{max}} = 1.82 \text{ e Å}^{-3}$

2114 reflections $\Delta \rho_{min} = -1.19 \text{ e Å}^{-3}$

128 parameters Extinction correction: none

Special details

Geometry. All s.u.'s (except the s.u. in the dihedral angle between two l.s. planes) are estimated using the full covariance matrix. The cell s.u.'s are taken into account individually in the estimation of s.u.'s in distances, angles and torsion angles; correlations between

s.u.'s in cell parameters are only used when they are defined by crystal symmetry. An approximate (isotropic) treatment of cell s.u.'s is used for estimating s.u.'s involving l.s. planes.

Refinement. Refinement of F^2 against ALL reflections. The weighted R-factor wR and goodness of fit S are based on F^2 , conventional R-factors R are based on F, with F set to zero for negative F^2 . The threshold expression of $F^2 > 2\sigma(F^2)$ is used only for calculating R-factors(gt) etc. and is not relevant to the choice of reflections for refinement. R-factors based on F^2 are statistically about twice as large as those based on F, and R- factors based on ALL data will be even larger.

Fractional atomic coordinates and isotropic or equivalent isotropic displacement parameters (\mathring{A}^2)

	x	y	Z	$U_{\rm iso}*/U_{\rm eq}$	Occ. (<1)
Ag	0.05548 (4)	0.51279 (13)	0.09923 (9)	0.0397 (4)	0.878 (4)
Nb1	0.13913 (3)	0.05530 (10)	0.00408 (6)	0.0156 (2)	
Nb2	0.36226 (3)	0.43048 (10)	0.28691 (6)	0.0155 (2)	
P	0.40291 (10)	0.1092 (3)	0.1376 (2)	0.0223 (5)	
S1	0.04299 (12)	0.4143 (4)	0.3739 (3)	0.0399 (7)	
S2	0.06326 (9)	0.1230 (3)	0.56424 (18)	0.0208 (5)	
S3	0.06437 (10)	0.1186 (3)	0.12793 (19)	0.0240 (5)	
S4	0.15627 (9)	0.1535 (3)	0.35711 (17)	0.0186 (5)	
S5	0.21371 (10)	0.1063 (3)	0.14306 (18)	0.0243 (5)	
S6	0.28921 (9)	0.4481 (3)	0.13057 (18)	0.0204 (5)	
S7	0.28941 (10)	0.3584 (3)	0.40463 (19)	0.0245 (5)	
S8	0.34993 (11)	0.1158 (3)	0.00528 (19)	0.0276 (6)	
S9	0.36001 (11)	0.0935 (3)	0.2684 (2)	0.0257 (5)	
S10	0.43518 (9)	0.3553 (3)	0.15040 (17)	0.0185 (5)	

Atomic displacement parameters (\mathring{A}^2)

	U^{11}	U^{22}	U^{33}	U^{12}	U^{13}	U^{23}
Ag	0.0377 (6)	0.0321 (6)	0.0492 (7)	-0.0076 (4)	0.0037 (5)	-0.0051 (5)
Nb1	0.0138 (4)	0.0169 (4)	0.0160 (4)	-0.0008(3)	0.0009(3)	0.0006(3)
Nb2	0.0131 (4)	0.0155 (4)	0.0178 (4)	0.0003(3)	0.0014(3)	0.0006(3)
P	0.0228 (13)	0.0150 (12)	0.0295 (13)	-0.0008 (10)	0.0049 (10)	0.0013 (10)
S1	0.0326 (15)	0.0216 (13)	0.066(2)	-0.0078 (12)	0.0070 (14)	0.0056 (13)
S2	0.0156 (11)	0.0233 (12)	0.0234 (12)	0.0028 (9)	0.0012 (9)	-0.0009 (10)
S3	0.0205 (12)	0.0256 (13)	0.0260 (12)	0.0066 (10)	0.0025 (9)	0.0025 (10)
S4	0.0193 (11)	0.0152 (11)	0.0212 (11)	-0.0010 (9)	0.0010 (9)	-0.0012 (9)
S5	0.0189 (11)	0.0324 (14)	0.0211 (12)	-0.0080 (10)	-0.0006 (9)	0.0029 (10)
S6	0.0148 (11)	0.0253 (12)	0.0209 (11)	-0.0024 (9)	0.0011 (9)	0.0001 (9)
S7	0.0200 (12)	0.0309 (13)	0.0230 (12)	-0.0063 (10)	0.0034 (9)	-0.0016 (10)
S8	0.0395 (15)	0.0215 (13)	0.0210 (12)	-0.0060 (11)	-0.0016 (11)	-0.0024 (10)
S9	0.0313 (13)	0.0191 (12)	0.0274 (13)	-0.0017 (10)	0.0068 (10)	0.0018 (10)
S10	0.0156 (10)	0.0182 (11)	0.0214 (12)	-0.0005 (9)	0.0004 (9)	0.0022 (9)

Geometric parameters (Å, °)

Ag—S1 ⁱ	2.536 (3)	Nb2—S3 ⁱⁱ	2.476 (3)
Ag—S9 ⁱⁱ	2.620(3)	Nb2—S7	2.479 (3)

:::	0.055 (0)	::	2 700 (2)
Ag—S2 ⁱⁱⁱ	2.875 (3)	Nb2—S5 ⁱⁱ	2.508 (2)
Ag—S8 ^{iv}	2.916 (3)	Nb2—S2 ^{viii}	2.551 (3)
Ag—S1 ⁱⁱⁱ	2.965 (4)	Nb2—S4 ⁱⁱ	2.558 (2)
Ag—S3	3.091 (3)	Nb2—S6	2.569 (3)
Ag—P ^{iv}	3.440 (3)	Nb2—S9	2.630(3)
Ag — Ag^{V}	3.549 (2)	Nb2—S10	2.656 (2)
Ag—P ⁱⁱ	3.552 (3)	Nb1—Nb2 ^{vii}	2.8800 (13)
Ag—Nb1	4.3137 (15)	Nb1—P ^{iv}	3.298 (3)
Nb1—S5	2.462 (2)	Nb1—Nb2 ^{iv}	3.7702 (15)
Nb1—S2 ^{vi}	2.466 (2)	Nb2—Nb1 ⁱⁱ	2.8800 (13)
Nb1—S7 ^{vii}	2.518 (3)	Nb2—Nb1 ^{iv}	3.7702 (15)
Nb1—S6 ^{iv}	2.551 (2)	P—S1 ^{vii}	2.009 (4)
Nb1—S3	2.554 (3)	P—S8	2.048 (4)
Nb1—S4 ^{vi}	2.562 (2)	P—S9	2.059 (4)
Nb1—S8 ^{iv}	2.573 (3)	P—S10	2.065 (3)
Nb1—S10 ^{iv}	2.659 (2)		
S1 ⁱ —Ag—S9 ⁱⁱ	131.39 (11)	S7—Nb2—S5 ⁱⁱ	47.83 (9)
S1 ⁱ —Ag—S2 ⁱⁱⁱ	113.11 (9)	S3 ⁱⁱ —Nb2—S2 ^{viii}	48.11 (8)
S9 ⁱⁱ —Ag—S2 ⁱⁱⁱ	79.08 (8)	S7—Nb2—S2 ^{viii}	89.05 (8)
S1 ⁱ —Ag—S8 ^{iv}	136.99 (9)	S5 ⁱⁱ —Nb2—S2 ^{viii}	107.42 (8)
S9 ⁱⁱ —Ag—S8 ^{iv}	78.25 (8)	S3 ⁱⁱ —Nb2—S4 ⁱⁱ	89.92 (8)
S2 ⁱⁱⁱ —Ag—S8 ^{iv}	101.58 (7)	S7—Nb2—S4 ⁱⁱ	120.91 (8)
S1 ⁱ —Ag—S1 ⁱⁱⁱ	100.08 (10)	S5 ⁱⁱ —Nb2—S4 ⁱⁱ	78.94 (8)
S9 ⁱⁱ —Ag—S1 ⁱⁱⁱ	127.42 (9)	S2 ^{viii} —Nb2—S4 ⁱⁱ	136.85 (8)
S2 ⁱⁱⁱ —Ag—S1 ⁱⁱⁱ	70.06 (8)	S3 ⁱⁱ —Nb2—S6	137.11 (9)
S8 ^{iv} —Ag—S1 ⁱⁱⁱ	68.05 (8)	S7—Nb2—S6	91.58 (8)
S1 ⁱ —Ag—S3	74.93 (8)	S5 ⁱⁱ —Nb2—S6	77.74 (8)
S9 ⁱⁱ —Ag—S3	96.70 (8)	S2 ^{viii} —Nb2—S6	173.34 (8)
S2 ⁱⁱⁱ —Ag—S3	171.84 (8)	S4 ⁱⁱ —Nb2—S6	47.42 (8)
S8 ^{iv} —Ag—S3	70.59 (7)	S3 ⁱⁱ —Nb2—S9	129.58 (9)
S1 ⁱⁱⁱ —Ag—S3	107.95 (8)	S7—Nb2—S9	79.70 (8)
S1 ⁱ —Ag—P ^{iv}	112.54 (9)	S5 ⁱⁱ —Nb2—S9	124.43 (9)
S9 ⁱⁱ —Ag—P ^{iv}	112.55 (8)	S2 ^{viii} —Nb2—S9	85.10 (8)
S2 ⁱⁱⁱ —Ag—P ^{iv}	95.91 (7)	S4 ⁱⁱ —Nb2—S9	127.37 (8)
S8 ^{iv} —Ag—P ^{iv}	36.42 (7)	S6—Nb2—S9	88.49 (8)
S1 ⁱⁱⁱ —Ag—P ^{iv}	35.58 (7)	S3 ⁱⁱ —Nb2—S10	86.73 (8)
S3—Ag—P ^{iv}	79.17 (7)	S7—Nb2—S10	153.96 (9)
S1 ⁱ —Ag—Ag ^v	55.36 (8)	S5 ⁱⁱ —Nb2—S10	154.21 (8)
S9 ⁱⁱ —Ag—Ag ^v	168.61 (8)	S2 ^{viii} —Nb2—S10	90.51 (8)
S2 ⁱⁱⁱ —Ag—Ag ^v	89.75 (6)	S4 ⁱⁱ —Nb2—S10	75.33 (7)
$S8^{iv}$ — Ag — Ag^v	102.14 (7)	S6—Nb2—S10	86.02 (8)
50 115 11g		<u>_</u> 510	00.02 (0)

S1 ⁱⁱⁱ —Ag—Ag ^v	44.72 (6)	S9—Nb2—S10	74.33 (8)
S3—Ag—Ag ^v	94.11 (6)	S3 ⁱⁱ —Nb2—Nb1 ⁱⁱ	56.36 (6)
S5—Nb1—S2 ^{vi}	111.70 (8)	S7—Nb2—Nb1 ⁱⁱ	55.44 (6)
S5—Nb1—S7 ^{vii}	47.90 (9)	S5 ⁱⁱ —Nb2—Nb1 ⁱⁱ	53.84 (6)
S2 ^{vi} —Nb1—S7 ^{vii}	90.11 (8)	S2 ^{viii} —Nb2—Nb1 ⁱⁱ	53.60 (6)
S5—Nb1—S6 ^{iv}	90.67 (8)	S4 ⁱⁱ —Nb2—Nb1 ⁱⁱ	116.29 (6)
S2 ^{vi} —Nb1—S6 ^{iv}	140.23 (9)	S6—Nb2—Nb1 ⁱⁱ	131.50 (6)
S7 ^{vii} —Nb1—S6 ^{iv}	80.99 (8)	S9—Nb2—Nb1 ⁱⁱ	114.81 (6)
S5—Nb1—S3	90.87 (8)	S10—Nb2—Nb1 ⁱⁱ	139.55 (6)
S2 ^{vi} —Nb1—S3	48.16 (8)	S3 ⁱⁱ —Nb2—Nb1 ^{iv}	112.00 (6)
S7 ^{vii} —Nb1—S3	107.97 (8)	S7—Nb2—Nb1 ^{iv}	132.44 (6)
S6 ^{iv} —Nb1—S3	168.99 (8)	S5 ⁱⁱ —Nb2—Nb1 ^{iv}	113.61 (6)
S5—Nb1—S4 ^{vi}	119.57 (9)	S2 ^{viii} —Nb2—Nb1 ^{iv}	135.05 (6)
S2 ^{vi} —Nb1—S4 ^{vi}	92.77 (8)	S4 ⁱⁱ —Nb2—Nb1 ^{iv}	42.61 (5)
S7 ^{vii} —Nb1—S4 ^{vi}	79.56 (8)	S6—Nb2—Nb1 ^{iv}	42.40 (5)
S6 ^{iv} —Nb1—S4 ^{vi}	47.56 (8)	S9—Nb2—Nb1 ^{iv}	86.53 (6)
S3—Nb1—S4 ^{vi}	138.99 (8)	S10—Nb2—Nb1 ^{iv}	44.85 (5)
S5—Nb1—S8 ^{iv}	78.70 (9)	Nb1 ⁱⁱ —Nb2—Nb1 ^{iv}	158.61 (3)
S2 ^{vi} —Nb1—S8 ^{iv}	130.83 (9)	S1 ^{vii} —P—S8	108.46 (17)
S7 ^{vii} —Nb1—S8 ^{iv}	123.89 (9)	S1 ^{vii} —P—S9	112.81 (17)
S6 ^{iv} —Nb1—S8 ^{iv}	84.30 (8)	S8—P—S9	111.87 (16)
S3—Nb1—S8 ^{iv}	85.31 (9)	S1 ^{vii} —P—S10	117.65 (16)
S4 ^{vi} —Nb1—S8 ^{iv}	124.80 (8)	S8—P—S10	104.24 (14)
S5—Nb1—S10 ^{iv}	155.38 (9)	S9—P—S10	101.46 (14)
S2 ^{vi} —Nb1—S10 ^{iv}	85.32 (8)	Nb1 ^{ix} —S2—Nb2 ^{viii}	70.04 (7)
S7 ^{vii} —Nb1—S10 ^{iv}	154.11 (8)	$S3^{ix}$ — $S2$ — Ag^x	146.81 (13)
S6 ^{iv} —Nb1—S10 ^{iv}	86.32 (8)	S2 ^{vi} —S3—Nb2 ^{vii}	67.87 (10)
S3—Nb1—S10 ^{iv}	87.75 (8)	S2 ^{vi} —S3—Nb1	63.68 (9)
S4 ^{vi} —Nb1—S10 ^{iv}	75.23 (7)	Nb2 ^{vii} —S3—Nb1	69.84 (7)
S8 ^{iv} —Nb1—S10 ^{iv}	76.69 (8)	S2 ^{vi} —S3—Ag	149.39 (13)
S5—Nb1—Nb2 ^{vii}	55.35 (6)	Nb2 ^{vii} —S3—Ag	132.70 (10)
S2 ^{vi} —Nb1—Nb2 ^{vii}	56.36 (6)	Nb1—S3—Ag	99.22 (8)
S7 ^{vii} —Nb1—Nb2 ^{vii}	54.18 (6)	S6 ^{vii} —S4—Nb2 ^{vii}	66.56 (9)
S6 ^{iv} —Nb1—Nb2 ^{vii}	134.56 (6)	S6 ^{vii} —S4—Nb1 ^{ix}	65.96 (9)
S3—Nb1—Nb2 ^{vii}	53.80 (6)	Nb2 ^{vii} —S4—Nb1 ^{ix}	94.84 (8)
S4 ^{vi} —Nb1—Nb2 ^{vii}	120.09 (6)	S7 ^{vii} —S5—Nb1	67.50 (10)
S8 ^{iv} —Nb1—Nb2 ^{vii}	112.84 (6)	S7 ^{vii} —S5—Nb2 ^{vii}	65.33 (10)
S10 ^{iv} —Nb1—Nb2 ^{vii}	137.39 (6)	Nb1—S5—Nb2 ^{vii}	70.81 (7)
S5—Nb1—Nb2 ^{iv}	132.20 (6)	S4 ⁱⁱ —S6—Nb1 ^{iv}	66.48 (9)
S2 ^{vi} —Nb1—Nb2 ^{iv}	112.84 (6)	S4 ⁱⁱ —S6—Nb2	66.02 (9)
S7 ^{vii} —Nb1—Nb2 ^{iv}	115.85 (6)	Nb1 ^{iv} —S6—Nb2	94.83 (8)

S6 ^{iv} —Nb1—Nb2 ^{iv}	42.76 (6)	S5 ⁱⁱ —S7—Nb2	66.84 (10)
S3—Nb1—Nb2 ^{iv}	132.49 (6)	S5 ⁱⁱ —S7—Nb1 ⁱⁱ	64.60 (10)
S4 ^{vi} —Nb1—Nb2 ^{iv}	42.55 (5)	Nb2—S7—Nb1 ⁱⁱ	70.38 (7)
S8 ^{iv} —Nb1—Nb2 ^{iv}	85.19 (6)	P—S8—Nb1 ^{iv}	90.34 (11)
S10 ^{iv} —Nb1—Nb2 ^{iv}	44.79 (5)	P—S9—Nb2	90.49 (11)
Nb2 ^{vii} —Nb1—Nb2 ^{iv}	161.96 (3)	P—S10—Nb2	89.62 (11)
P ^{iv} —Nb1—Nb2 ^{iv}	56.18 (5)	P—S10—Nb1 ^{iv}	87.61 (11)
S3 ⁱⁱ —Nb2—S7	111.78 (8)	Nb2—S10—Nb1 ^{iv}	90.36 (7)
S3 ⁱⁱ —Nb2—S5 ⁱⁱ	91.64 (9)		

Symmetry codes: (i) -x, y, -z+1/2; (ii) -x+1/2, y+1/2, -z+1/2; (iii) x, -y+1, z-1/2; (iv) -x+1/2, -y+1/2, -z; (v) -x, -y+1, -z; (vi) x, -y, z-1/2; (vii) -x+1/2, y-1/2, -z+1/2; (viii) -x+1/2, -z+1/2; (viii) -x+1/2; (viiii) -x+1/2; (viii) -x+1/2; (viiii) -x+1/2; (viii) -x+1/2; (viiii) -x+1/2; (viii) -x+1/2; (viii) -

Fig. 1

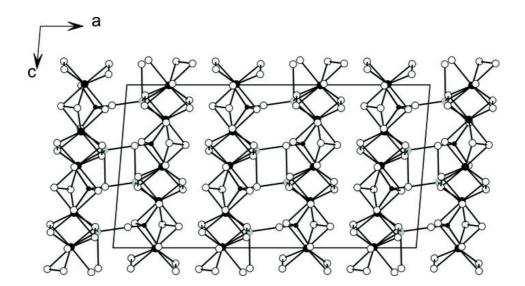


Fig. 2

